THE EFFECTS OF FE-DOPING ON MECHANICAL PROPERTIES IN Bi$_{1.6}$Pb$_{0.4}$Sr$_2$Ca$_2$Cu$_{3-x}$Fe$_x$O$_y$ SUPERCONDUCTOR

(Kesan Dop Ferum Terhadap Sifat Superkonduktor Bi$_{1.6}$Pb$_{0.4}$Sr$_2$Ca$_2$Cu$_{3-x}$Fe$_x$O$_y$)

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Abstract

Bi-2223 superconductors doped with Fe$_2$O$_3$ at Cu-site were prepared in bulk form using high purity oxide powders via solid state reaction technique with intermediate grinding. The series of Fe stoichiometric ratio ($x = 0.00, 0.02, 0.04, 0.06, 0.08$ and $0.10$) are systematically added to Bi$_{1.6}$Pb$_{0.4}$Sr$_2$Ca$_2$Cu$_{3-x}$Fe$_x$O$_y$ system to study the effect of Fe doping on mechanical and superconducting system properties. Vickers hardness, electrical resistivity, X-ray diffraction and Field Emission Scanning Electron Microscopy have been carried out to assess the effects of Fe doping. These measurements indicate that the Fe doping decreased the critical temperature and deteriorated formation of high $T_c$ phase, compared to undoped sample. Grain size of the samples were also reduced by increasing the amount of Fe and the mechanical properties were found to be escalate with the increasing of Fe concentration.

Keywords: BSCCO superconductor, Fe-doped, critical current density, X-ray diffraction, microhardness

Abstract

Superkonduktor Bi-2223 dop Fe$_2$O$_3$ dikawasan kuprum telah dihasilkan secara pukal menggunakan serbuk oksida tulin dengan kisaran perantara melalui kaedah tindak balas keadaan pepejal. Nisbah stokiometri bersiri ($x = 0.00, 0.02, 0.04, 0.06, 0.08$ and $0.10$) ditambah secara sistematik terhadap sistem Bi$_{1.6}$Pb$_{0.4}$Sr$_2$Ca$_2$Cu$_{3-x}$Fe$_x$O$_y$ bagi mengkaji kesan dop Fe terhadap sifat mekanikal dan sistem superkonduktor. Ujian ketahanan Vickers, kerintangan arus, Sinar X, difraksi X-ray dan Mikroskop Elektron Penananan telah dilakukan untuk menilai kesan dop Fe. Penyelidikan ini menunjukkan bahawa kesan dop yang dilakukan mengurangkan genting dan merencatkan pembentukan fasa suhu tinggi berbanding sampel tanpa dop. Siai butiran sampel turut mengecil dengan penambahan jumlah kehadiran Fe dan sifat mekanikal didapati meningkat seiring dengan penambahan Fe.

Kata kunci: Superkonduktor BSCCO, dop-Fe, ketumpatan arus, pembelauan sinar-X, ketahanan mikro

Introduction

Bi-Sr-Ca-Cu-O compound has been extensively studied since the discovery of high superconducting temperature [1]. This system has been investigated by many research groups concerning the preparation, superconducting properties, effect of doping as well as the structure of these compounds [2-6]. Among the BSCCO system, Bi-2223 is both fundamental and technological subject of interest because of its higher critical temperature, $T_c$ [7]. The grain
boundaries were essential as current transport factor in the polycrystalline form. Optimal sintering time is important to enhance the superconducting volume fraction and grains connectivity thus increases the critical sample current density. Short sintering time can produce smaller grain size, resulting in weak connectivity between grains. However, prolonging the sintering time may lead to phase decomposition, decreasing the superconducting phases and weakening the weak links. Bi-2223 is amenable for possible practical application as a conductor due to their large critical current density [8, 9].

However, it is difficult to obtain higher critical current density \( J_c \), which is significant for promising superconductor system. The selective substitution studies are essential and interesting for superconductivity research area in order to improve the critical current density. Elemental substitution is one of the important roles to enhance the \( T_c \) and \( J_c \) by improving the phase formation or microstructure of the system [10, 11]. Practical applications, in terms of the BSCCO mechanical properties were also important for the studies since high temperature superconductors are generally restricted due to their brittleness in nature. The various substitutions have been investigated in order to improve the formation and stability of the 2223 phases and the most effective one is partial substitution of trivalent Bi with Pb. The 3d elements also embedded for substitution in Bi-2223 system.

In this paper, the effect of 3d elements (Fe) that has been substituted into Bi (Pb)-2223 superconductor through solid state reaction method was presented. The structural and transport properties of Bi\(_{1-x}\)Pb\(_x\)Sr\(_2\)Ca\(_2\)Cu\(_{3-x}\)Fe\(_x\)O\(_y\) were characterized using X-ray diffraction (XRD) and electrical resistivity measurement respectively while mechanical properties was unveiled using Vickers hardness.

**Materials and Methods**
The high-\( T_c \) superconducting samples with nominal composition Bi\(_{1-x}\)Pb\(_x\)Sr\(_2\)Ca\(_2\)Cu\(_{3-x}\)Fe\(_x\)O\(_y\) (where \( x = 0.00, 0.02, 0.04, 0.06, 0.08 \) and 0.10) ceramic superconductor were prepared using the solid state reaction method. The Bi\(_3\)O\(_3\), PbO, SrCO\(_3\), CaCO\(_3\), Fe\(_2\)O\(_3\) and CuO powders with 99.9% purity were used. The powders were weighed according to their stoichiometry ratio. The powders were ball-milled together with absolute ethanol in alumina crucible for 24 hours. After that, the homogenous mixture of the powders was dried out in the oven at the 120 °C for 6 hours. The powders were ground and pre-calcined at 800 °C for 15 hours. Then the mixture of the powder was again calcined at 820 °C for 15 hours to remove all oxides and carbonates. After that, the powders were pressed into pellet of 2 grams by applying a 30 MPa pressure. Finally, the pellets were sintered at 850 °C for 48 hours to form a high-\( T_c \) phase BSCCO superconductor. Results for \( T_c \), \( I_c \), and \( J_c \) of the samples were measured by using four point probe machine. XRD analysis of the samples was carried out by using Cu (K\(_\alpha\)) radiation. The surface morphologies of the samples were studies using FESEM machine, and the mechanical properties of the compounds have been investigated using Vickers hardness. All samples were denoted as sample A, B, C, D, E, and F for \( x = 0.00, 0.02, 0.04, 0.06, 0.08 \), and 0.10 respectively.

**Results and Discussion**
The normalized resistance at room temperature as a function of temperature between 30 K and 300 K with various concentration of Fe to Bi-2223 powder are shown in Figure 1. The curve indicated a metallic behaviour for all samples at normal state with a single leap of superconducting transition. The zero resistance temperature of the samples decreased with the addition of more dopant concentration into the system. Sample of Fe-doped with \( x = 0.10 \) shows that there is no any transition to the superconducting state due to no \( T_c \) zero. The highest \( T_c \) zero was showed in pure sample is 93 K and decreases to 57 K for sample \( x = 0.08 \).

Figure. 2 shows the XRD patterns of samples. All samples consisted of the Bi-2223 and Bi-2212 phase with the increasing of Bi-2212 phase induction. No peaks belongs to Fe were detected, which implied that the Fe had incorporated into the crystal structure subjected to the low amount of Fe.
Figure 1. Normalized resistance as a function of temperature of the samples

Figure 2. X-ray diffraction pattern of samples. The peaks indexed represent Bi-2212

Figure 3 shows the Vickers microhardness, Hv versus the concentration of Fe, \( x \) for all samples and the value of the Hv was tabulated in Table 1. It showed that the mechanical properties of the compounds were found to be linearly increased with the increasing of doping in Bi-2223 superconductors [12-15]. A sudden dropped was observed when \( x = 0.08 \) and this behaviour is might be due to the contribution of weak grain boundaries in the sample.
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Figure 3. Vickers microhardness, Hv versus concentration of Fe, x

Table 1. Critical temperature ($T_c$), critical current density ($J_c$), lattice parameter $a$, $b$ and $c$, volume fraction, and Vickers hardness (Hv) values for the samples

<table>
<thead>
<tr>
<th>Sample</th>
<th>$T_c$ zero (K)</th>
<th>$J_c$ (A/cm$^2$)</th>
<th>$% V_{2223}$</th>
<th>a</th>
<th>b</th>
<th>c</th>
<th>Hv (GPa)</th>
</tr>
</thead>
<tbody>
<tr>
<td>A</td>
<td>93</td>
<td>4.8728</td>
<td>78.46</td>
<td>5.3891</td>
<td>5.4036</td>
<td>36.5143</td>
<td>0.529</td>
</tr>
<tr>
<td>B</td>
<td>89</td>
<td>4.6180</td>
<td>75.06</td>
<td>5.3891</td>
<td>5.4036</td>
<td>36.5143</td>
<td>0.658</td>
</tr>
<tr>
<td>C</td>
<td>83</td>
<td>2.6225</td>
<td>66.13</td>
<td>5.3891</td>
<td>5.3434</td>
<td>33.9262</td>
<td>0.989</td>
</tr>
<tr>
<td>D</td>
<td>81</td>
<td>2.4026</td>
<td>29.64</td>
<td>5.3891</td>
<td>5.3582</td>
<td>33.8002</td>
<td>1.293</td>
</tr>
<tr>
<td>E</td>
<td>57</td>
<td>0.6197</td>
<td>24.48</td>
<td>5.3891</td>
<td>6.6780</td>
<td>30.6833</td>
<td>0.755</td>
</tr>
<tr>
<td>F</td>
<td>-</td>
<td>-</td>
<td>26.59</td>
<td>5.3891</td>
<td>5.5929</td>
<td>30.6201</td>
<td>0.691</td>
</tr>
</tbody>
</table>

The FESEM micrographs of all samples in Figure 4 consisted of plate-like grains with randomly distributed as the typical grain structure of the high-$T_c$ phase (Bi-2223). It is observed that the grain connectivity is worsened greatly with increasing Fe concentration. The surface of the Fe-free sample is more uniform with better grains alignment. FESEM images of $x = 0.00$ and $x = 0.02$ show better crystallinity in comparison with the remaining two samples. Sample $x = 0.10$ has the worst appearance among these six samples. These results indicate that the surface morphology of the sample is worsened with increasing of Fe substitution.
Figure 4. FESEM micrograph for the cross-section of the samples for a) $x = 0.00$, b) $x = 0.02$, c) $x = 0.04$, d) $x = 0.06$, e) $x = 0.08$ and f) $x = 0.10$

Conclusion
The influence of Fe doping on mechanical properties in Bi$_{1.6}$Pb$_{0.4}$Sr$_2$Ca$_2$Cu$_{3-x}$Fe$_x$O$_y$ system has been investigated. The results generally showed that the doping of Fe increased the mechanical properties of the samples until $x = 0.06$, and the value dropped when $x = 0.08$. The doping of Fe also does not favour the formation of high $T_c$ phase. The samples exhibit the decrease values of $T_c$ and $J_c$ after further addition of dopant concentrations. XRD analysis showed that the volume fraction of the 2223 phases was decreased and the volume fraction of the 2212 phases was dominant with increasing of Fe doping. These measurements indicate that Fe doping decreased the critical temperature and degrade the formation of high-$T_c$ phase of high temperature superconductor (HTS) compared with the undoped sample.

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